Materials LAS

Tin sulfide chalcogel derived SnS^x for CO² electroreduction

Published as part of the Virtual Special Issue "Mercouri G. Kanatzidis at 65" Jingwen Bai^{1,2†}, Lijun Yang^{2†}, Yuanyuan Zhang^{1,2}, Xiaofu Sun³ and Jian Liu^{1,2[*](http://orcid.org/0000-0002-9684-339X)}i

A new class of aerogels based exclusively on metal chalcogenide frameworks have been developed, opening up a series of novel properties and applications. Further expanding the application of such chalcogels in electrocatalytic CO² reduction is of significance for mitigating the rise of atmospheric CO² concentration. Herein, the tin sulfide chalcogel was employed as a pre-catalyst for the construction of efficient electrocatalysts for CO² reduction. SnS0.09 and SnS0.55 supported on carbon cloth (SnS0.09/CC and SnS0.55/CC) were obtained with different amounts of sulfur by cyclic voltammetry activation of the tin sulfide chalcogel at different potential intervals. Compared with SnS0.09/CC, SnS0.55/CC with higher S contents exhibited a higher formate Faraday efficiency of 93.1% at −1.1 V verse reversible hydrogen electrode, and the partial current density of formate was 28.4 mA/cm² . The difference in performance between SnS0.09/CC and

SnS0.55/CC could be attributed to the varying sulfur contents which could favor the formation of formate.

halcogels are a newly emerging class of highly por-
ous aerogels based exclusively on metal chalcogen-
ide frameworks.^[1-4] Chalcogels have the general
properties of metal chalcogenide compounds, such as unique halcogels are a newly emerging class of highly porous aerogels based exclusively on metal chalcogen-ide frameworks.^{[1-[4\]](#page-5-1)} Chalcogels have the general pore structures, suitable band gaps, semiconductor proper-ties and so on.^{[\[1,](#page-5-0)[5](#page-5-2)-7]} Due to the abundance of soluble chalcogenide clusters and metal ions, the properties of chalcogels could be tuned through carefully choosing the appropriate anions and cations.^{[\[1](#page-5-0),[5](#page-5-2)[,8](#page-5-4),[9\]](#page-6-0)} Besides, compared with metal chalcogenide compounds, the one-pot synthesis of chalcogels was usually accomplished at ambient conditions.^{[[6](#page-5-5)[,10,](#page-6-1)[11\]](#page-6-2)} These features allow the chalcogels to show potential applications in catalysis, gas separation and adsorption, environmental re-mediation, and so on.^{[[3,](#page-5-6)[4](#page-5-1),[8](#page-5-4),12-[18](#page-6-4)]}

Studies of the chalcogels in electrocatalytic applications have focused mainly on electrocatalytic hydrogen evolution reactions, $[19-21]$ $[19-21]$ $[19-21]$ $[19-21]$ whereas it is equally necessary to extend to

³ *Beijing National Laboratory for Molecular Sciences, CAS Key Laboratory of Colloid and Interface and Thermodynamics, CAS Research/Education Center for Excellence in Molecular Sciences, Institute of Chemistry, Chinese Academy of Sciences, Beijing 100190, China*

Received 4 July 2022; Accepted 3 August 2022; Published online †These authors contributed equally to this work.

 CO_2 reduction reaction (CO₂RR). As well known, with the combustion of fossil fuels, the continuous rise of atmospheric $CO₂$ concentration poses a serious threat to the environment and climate. In order to mitigate atmospheric $CO₂$ concentration, electrocatalytic $CO₂RR$ is an appealing and sustainable strategy, as the electricity can be provided by abundant and renewable energy sources.[[22](#page-6-7)–[24](#page-6-8)] The feasibility of chalcogels in electrocatalytic $CO₂RR$ has been demonstrated by Kanatzidis et al. using biomimetic chalcogels comprised of $\textsf{Na}_4\textsf{Sn}_2\textsf{S}_6$ 14H₂O and $(\textsf{Ph}_4\textsf{P})_2[\textsf{Fe}_4\textsf{S}_4\textsf{Cl}_4]$.^{[[11](#page-6-2)]} However, the active sites of such catalysts were not very clear. Therefore, further exploring the study of chalcogels in electrocatalytic $CO₂RR$ and establishing the structure-activity relationship are of great significance.

Among the reported chalcogels, the tin sulfide chalcogel (SnS_x) has a simple composition consisting of $Sn²⁺$ and [SnS₄]⁴⁻, which is more favorable for establishing the relationship between composition and reaction performance. Tin sulfide has been widely investigated in the field of elec-trocatalysis.^{[25-[30\]](#page-6-10)} In addition, tin-based materials have been widely employed as electrocatalysts for $CO₂RR$ to produce formate.[\[29,](#page-6-11)[31](#page-6-12)–[37](#page-6-13)] Among the various liquid products, formate is considered to be one of the most economically feasible products, which is widely used as an important raw material in the chemical industries and a promising carrier for hydro-gen storage.^{[[38](#page-6-14)[–41\]](#page-6-15)} Therefore, SnS_x chalcogel was tentatively employed as the starting material for construction of the efficient electrocatalyst for $CO₂RR$.

¹ College of Materials Science and Engineering, Qingdao University of Sci*ence and Technology, Qingdao 266042, China*

² *Qingdao Institute of Bioenergy and Bioprocess Technology, Chinese Academy of Sciences, Shandong Energy Institute, Qingdao 266101, China*

^{*} Corresponding author, E-mail: liujian@qibebt.ac.cn

[©] 2023 The Author(s).*Materials Lab* published by Lab Academic Press *Materials Lab* 2023, 2, 220046

In our study, we found that S_nS_x was unstable and underwent morphological and composition changes after cyclic voltammetry (CV) activation. Such changes prompted us to employ the chalcogel S_nS_x as a pre-catalyst for further study. The "real" catalysts after CV activation at different potential intervals were termed as $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$, respectively. According to the characterization of the energy-dispersive X-ray spectroscopy (EDS) and X-ray photoelectron spectroscopy (XPS), the S contents were found to be decreased for both $\text{SnS}_{0.09}/\text{CC}$ and $\text{SnS}_{0.55}/\text{CC}$ after CV activation, and the latter had higher S contents than the former counterpart. At −1.1 V verse reversible hydrogen electrode (vs. RHE), $SnS_{0.55}/CC$ demonstrated a higher formate Faraday efficiency of 93.1% and the formate partial current density of 28.4 mA/cm², surpassing those of SnS_{0.09}/CC. The phenomenon was attributed to the higher S contents in $SnS_{0.55}/CC$, which facilitated the formation of formate. The current study could shed light on the construction of metal sulfide electrocatalysts by in-situ electrochemical activation.

Materials and methods

Chemicals and materials

All the reagents were purchased commercially and used without further purification. Sodium sulfide nonahydrate $(Na₂S·9H₂O)$, methanol, ethanol, formamide, potassium bicarbonate (KHCO₃) were purchased from Sinopharm Chemical Reagent Co. Ltd.. Tin (IV) chloride pentahydrate (SnCl₄·5H₂O) was purchased from Shanghai Aladdin Biochemical Technology Co. Ltd.. Tin (II) acetate (Sn(OAc)₂) was purchased from Energy Chemical Co. Ltd.. Nafion 117 containing solution was purchased from Sigma Aldrich.

Synthesis of Na4SnS⁴ ·14H2O

 $Na₂S·9H₂O$ (120 mmol) was dissolved in 100 mL H₂O with stirring. Then, SnCl $_4$ ·5H $_2$ O (30 mmol) was dissolved in 5 mL H_2O and added dropwise to the Na₂S·9H₂O solution with continuous stirring. The solution was kept at 45 °C for 8 h and then added to 400 mL methanol. The flask was placed in a refrigerator for 48 h. Precipitated white crystal of $\mathsf{Na}_4\mathsf{SnS}_4\textnormal{\textsf{.14H}}_2\mathsf{O}$ was washed with EtOH and dried under vacuum for 24 h.

Synthesis of tin sulfide chalcogel (SnS^x)

Tin sulfide chalcogel (SnS_x) was synthesized by using a modification of the literature method as follows^{[[42](#page-6-16)]}: $\mathsf{Na}_4\mathsf{SnS}_4\mathsf{\cdot}$ 14H $_2$ O (0.2 mmol) and Sn(OAc) $_2$ (0.4 mmol) were dissolved in a total 8 mL of formamide in two separate vials. The $Sn(OAc)_2$ solution was added very slowly to the Na₄SnS₄·14H₂O solution with vigorous stirring. Small, yellowish-orange particles appeared upon the addition of a drop of $\mathsf{Sn}(\mathsf{OAc})_2$ solution, but they disappeared with shaking. Addition of the $Sn(OAc)_2$ was followed by vigorous shaking. Then, the vial was placed in an oil bath (60 $^{\circ}$ C) for 7 days. The remaining formamide was decanted and the gel was subsequently washed with water and ethanol over 3 days and finally dried through freeze-drying, resulting in the SnS_x powders.

Electrode preparation

10 mg of SnS_x, 3 mg of carbon black, and 50 μ L of Nafion solution (5 wt%) were dispersed in 1 mL of ethanol via sonica-

tion for 30 min to form a homogeneous ink. Then, the as-prepared catalyst ink was drop-cast onto a piece of cleaned carbon cloth with an area of 1×1 cm² to form the electrode with catalyst loading of 1 mg/cm². The prepared electrodes were dried under ambient environment.

Electrochemical activation of SnS^x in different potential regions

The as-obtained SnS_x electrodes were electro-activated by 20-segment consecutive CV scans through two different potential intervals in CO₂-saturated 0.5 M KHCO₃ with a scan rate of 50 mV/s. According to the quantitative analysis of XPS, the working electrode obtained by electrochemical activation from 0.16 V to -1.54 V vs. RHE was named SnS_{0.09}/CC. The working electrode obtained by electrochemical activation from 0.76 V to -1.54 V vs. RHE was named SnS_{0.55}/CC.

Electrochemical measurements

All electrochemical $CO₂$ reduction experiments was conducted in a three-electrode system in a gas-tight H-cell (separated by Nafion 117) containing 180 mL of 0.5 M KHCO₃ electrolyte at room temperature and under atmospheric pressure. $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$ were used directly as the working electrodes. Ag/AgCl (with saturated KCl as the filling solution) and platinum mesh were used as the reference and counter electrodes, respectively. The potentials were controlled by a CHI660E electrochemical station. Electrode potentials were converted to RHE scale according to the Nernst equation: on: on: on: $E_{RHE} = E_{Aq/AGCI} + 0.197 V + 0.0591 \times pH$. The pH values of N_{2} - and CO₂-saturated 0.5 M KHCO₃ electrolyte were 8.52 and 7.70, respectively. After $CO₂$ was purged into the KHCO₃ solution for at least 30 min to saturate the electrolyte, electroreduction of $CO₂$ was investigated for 1 h at each applied potential by the controlled potential electrolysis method. The average mass flowrate of CO₂ (99.999%) was controlled at 25 mL/min by a mass flow controller (YJ-700CF) at the inlet of the electrochemical cell. The gaseous products of $CO₂$ electrocatalytic reduction were monitored by an on-line gas chromatography (GC) (GC9790plus, Fuli Instruments) equipped with the flame ionization detector (FID) and thermal conductivity detector (TCD). H_2 and CO concentrations were analyzed by the TCD and FID detectors, respectively. The KHCO₃ solution after electrolysis was collected and analyzed on an ion chromatograph (863 Basic IC plus, Metrohm, Switzerland) equipped with a MetrosepA Supp 4-250/4.0 column to quantify liquid products (formate). Faradaic efficiencies (FE) of formate, CO and H₂ were calculated by the equation: FE (%) = nFz/Q, where n represents the total number of moles of formate, CO and $H₂$ production, which were measured by ion chromatograph and GC; F represented Faraday constant (96485); z represented the number of electrons required to form formate, CO and H_2 from CO₂; and the Q corresponded to the amount of cumulative charge in the process of CO_2 reduction, which was provided by the electrochemical workstation.

The linear scanning voltammetry (LSV) curves of $SnS_{0.09}/CC$ and SnS_{0.55}/CC were carried out in CO₂-saturated or N₂-saturated 0.5 M KHCO₃ with the scan rate of 5 mV/s based on the above electrolysis cell. Double-layer capacitance (Cdl) was determined by measuring the capacitive current associated with double-layer charging from the scan-rate dependence of CV. The CV ranged from 0.31 V to 0.41 V vs. RHE. The Cdl was estimated by plotting the *Δ*J (*Δ*J = j_a−j_c) at 0.36 V vs. RHE against the scan rates, in which the j_a and j_c were the anodic and cathodic current density, respectively. The slope was twice that of the Cdl values. The electrochemical impedance spectra of the $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$ were recorded at -0.8 V vs. RHE in CO₂-saturated 0.5 M KHCO₃ electrolyte with an amplitude of e frequency range is from 0.1 Hz to 100000 Hz.

Characterizations

Scanning electron microscopy (SEM) and EDS was performed on a JEOL JSM-6700F scanning electron microscope equipped with an energy dispersive X-ray spectroscopy detector. Transmission Electron Microscope (TEM) and High-Resolution Transmission Electron Microscopy (HRTEM) were carried out on a JEOL JEM-2100Plus transmission electron microscope with an acceleration voltage of 200 kV. Powder X-ray diffraction (XRD) patterns were recorded by using a Rigaku Ultima IV diffractometer (Cu K*α* radiation, *λ*=1.5418 Å) with an operating voltage of 40 kV and a current of 44 mA. XPS measurements were carried out on a Thermo Scientific K-Alpha system, with a 400 µm spot size, 50 eV pass energy and energy steps of 0.1 eV, and aluminum anode X-ray excitation.

Results

Preparation and characterization of tin sulfide chalcogel (SnS^x)

In the present study, the S_nS_x chalcogel was synthesized by the metathesis reaction between Na₄SnS₄·14H₂O and Sn(OAc)₂. As shown in Fig. S1a, formamide solutions of two reactants were slowly mixed, and the resultant solution was kept at 60 °C for gelation. The SnS_x powders were obtained after the [solvent](#page-2-0) exchange [and fr](#page-2-0)eeze drying (Fig. S1b and S1c). SEM [\(Fig. 1a](#page-2-0)) and TEM [\(Fig. 1b](#page-2-0)) images showed the formation of micron-sized particles by aggregation of nanoparticles. HRTEM image reflected that SnS_x powders were crystalline with an interplanar distance of lattice fringes of 0.34 nm, which ascri[bed to](#page-2-0) the (201) plane of SnS and the (101) plane of Sn₂S₃ ([Fig. 1c\)](#page-2-0). EDS spectra showed that Sn, S,

Na and O presented in the SnS_x powders (Fig. S2). The small amount of Na originated from the reaction feedstock residue and the small amount of O may be due to the exposure of the catalyst to air. As indicated by elemental mapping, S and Sn elements were distributed homogeneously among SnS_x particles (Fig. S3). SnS_y consisted of SnS and $Sn₂S₃$ according to the XRD pattern, and the sharp diffraction peaks indicated the crystallinity of SnS_x, which also corresponded to the results of HRTEM ([Fig. 1d\)](#page-2-0). XPS survey spectra further confirmed the chemical composition of SnS $_{\mathrm{x}}$ (Fig. S4). As evidenced by XPS spectra of Sn 3d, Sn presented in the form of Sn²⁺ (494.7) eV and 486.3 eV) and Sn^{4+} (495.5 eV and 487.1 eV).^{[[37](#page-6-13)[,43\]](#page-6-17)} Two characteristic peaks at 162.7 eV and 161.5 eV were assigned to S 2 $p_{1/2}$ and S 2 $p_{3/2}$ of SnS_x, respectively ([Fig. 1f](#page-2-0)).^{[\[30](#page-6-10)[,43\]](#page-6-17)}

Electrochemical activation of SnS^x in different potential regions

SnS_x powders were loaded on a 1 \times 1 cm² area of carbon cloth (SnS_x/CC), which was used as a working electrode for electrochemical $CO₂RR$. CV scans were usually carried out prior to the performance tests to activate the electrodes. We applied 10 cycles of CV scans before other tests (scan rate: 50 mV/s). Typical CV curves are shown in Fig. S5b. A reduction peak at -0.35 V vs. RHE and an oxidation peak at 0.25 V vs. RHE emerged in the second voltammetric cycle (this process was poorly defined in the first voltammetric cycle), su[gg](#page-6-18)[es](#page-6-19)ting the occurrence of reduction and oxidation for SnS_x ^{[\[44,](#page-6-18)[45](#page-6-19)]} By adopting different potential intervals for activation of the electrodes, SnS_x would undergo different redox processes to obtain electrocatalysts with different morphologies and components. According to the atom ratio of Sn and S obtained by XPS quantitative analysis (Fig. S9 and Table S2), the electrodes activated at two different potential intervals were named as SnS_{0.09}/CC (0.16 V~-1.54 V vs. RHE) and SnS_{0.55}/CC (0.76 V~−1.54 V vs. RHE), respectively (Fig. S5a and S5b). The electrode $\text{SnS}_{0.09}/\text{CC}$ underwent only reduction, and $SnS_{0.55}/CC$ underwent both reduction and oxidation. From the optical photographs in Fig. S5c-S5e, gray-white substances appeared on the surface of both electrodes after CV,

further confirming the transformation of substances in the CV process. $SnS_{0.09}/CC$ produced more gray-white materials than $SnS_{0.55}/CC$, indicating that different potential ranges of CV activation had effects on SnS_x conversion in electrochemical processes. The effect of activation using different potential intervals on the morphology of the electrocatalysts was confirmed by SEM. As shown by [Fig. 2a,](#page-3-0) S6a and S6b, the pre c atalyst SnS_x/CC consisted of numerous nanoparticles that were composed of smaller ones. The average size of $\mathsf{SnS_x/CC}$ particles was 69.3 \pm 6 nm. Compared to SnS_x/CC, larger particles with the average size of 186 ± 4 nm appeared on the $SnS_{0.09}/CC$ electrode. In contrast, the particles did not grow significantly on the $SnS_{0.55}/CC$ electrode, with the average sizeof 73.3 \pm 9 nm ([Fig. 2b,](#page-3-0) [2c](#page-3-0), and S6c-S6f). The EDS mapping patterns of Sn and S in the three electrodes reflected the different S contents (Fig. S7). This feature was further evident by the elemental content analysis of the EDS (Fig. S8 and Table S1). The Sn/S atomic ratios of SnS_x/CC, SnS_{0.09}/CC and $SnS_{0.55}/CC$ were $1/1.68$, $1/0.09$ and $1/0.47$, respectively, which were consistent with the XPS results. It indicated that the S content of both electrocatalysts decreased after CV activation, and $SnS_{0.55}/CC$ contained more S elements than $SnS_{0.09}/CC$. The above results could illustrate that different activation potential intervals in CV activation affect the morphology and composition of the electrocatalysts.

To further investigate the composition and structure of $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$, we carried out a series of characterizations including [XRD an](#page-3-0)d XPS measurements. As evidenced by XRD patterns ([Fig. 2d\)](#page-3-0), $\text{SnS}_{0.09}/\text{CC}$ was consisted of metallic Sn, SnS and Sn₂S₃, indicating the reduction of SnS_x to metallic Sn occurred at a potential interval (0.16 V~−1.54 V) vs. RHE) with a small amount of unconverted SnS_x . In contrast, $SnS_{0.55}/CC$ just showed peaks with weak intensities assigning to Sn, SnS and Sn $_2$ S $_3$, respectively. The reason was ascribed to the fact that S_nS_x constantly underwent reduction and oxidation during electrochemical activation in the potential region of 0.76~−1.54 V vs. RHE. As evidenced by the Sn 3*d* XPS spectra of the three electrodes, $\text{SnS}_{0.09}/\text{CC}$ had the appearance of

the peak of Sn⁰, whereas this phenomenon was not observed for SnS_{0.55}/CC ([Fig. 2e](#page-3-0)).^{[\[37,](#page-6-13)[46\]](#page-6-20)} [Fig. 2f](#page-3-0) showed the S 2p XPS spectra of the three electrodes. Two characteristic peaks at 168.4 eV and 169.8 eV were assigned to sulfate/nafion.[[47](#page-6-21)[,48\]](#page-6-22) Two peaks at 162.7 eV and 161.5 eV attributed to S $2p_{1/2}$ and S 2 $p_{3/2}$ of SnS_x, respectively. [Fig. 2e](#page-3-0) and [2f](#page-3-0) also reflected that peaks shift of Sn and S in the XPS for $SnS_{0.09}/CC$ and $SnS_{0.55}/CC.$ Relative to $SnS_x/CC,$ the XPS peaks of Sn for $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$ shifted to lower binding energies, whereas the XPS peaks of S shifted to higher binding energies, which could be attributed to the electron transfer from S to Sn. The different degree of peaks shift could further infer that the different potential intervals in CV activation affected the electron transfer between Sn and S, which changed the electronic structure of the Sn sites and thus may influence the in[teraction b](#page-4-0)etween catalysts and CO_2 .^{[\[29\]](#page-6-11)}

[Scheme 1](#page-4-0) demonstrated the in-situ conversion processes of the pre-catalyst SnS_x/CC through different potential intervals in electrochemical activation. The CV curves of $\text{SnS}_{0.99}/\text{CC}$ showed only one reduction peak at −0.35 V vs. RHE, indicating the occurrence of only reduction reactions. Most of the S_nS_x was converted to metallic Sn, and a very small amount of S remained. In this conversion, the nanoparticles on the $SnS_{0.99}/CC$ electrode became larger. For $SnS_{0.55}/CC$, SnS_x was first converted to metallic Sn and S ions entered the electrolyte before the oxidation process occurred with S ions. During electrochemical activation, $SnS_{0.55}/CC$ has been following this redox process. Thus, compared to $\text{SnS}_{0.99}/\text{CC}$, $\text{SnS}_{0.55}/\text{CC}$ contained more S and smaller particles. Through the selection of different potential intervals in CV activation, the precatalyst transformed to two electrocatalysts with different particle sizes and composition, which was especially reflected in the atom ratio of Sn to S. The thus obtained catalysts feature different active sites, which were used in the electrocatalytic CO₂RR.

Electrochemical CO2RR performance of the SnS0.09/CC and SnS0.55/CC

Electrocatalytic CO₂RR performance of the SnS_{0.09}/CC and

Fig. 2 SEM images of **a** SnS_x/CC, **b** SnS_{0.09}/CC and **c** SnS_{0.55}/CC; **d** XRD patterns, **e** Sn 3d XPS spectra and **f** S 2p XPS spectra of SnS_x/CC, $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$.

 $SnS_{0.55}/CC$ samples were evaluated in a gas-tight H-cell containing 180 mL of CO₂-saturated 0.5 M KHCO₃ electrolyte, which was separated by a proton exchange membrane (Fig. S10). [Fig. 3a](#page-4-1) showed LSV curves of the $\text{SnS}_{0.09}/\text{CC}$ and SnS $_{0.55}$ /CC in CO₂- and N₂-saturated 0.5 M KHCO₃. As shown in the inset of [Fig. 3a](#page-4-1), the reduction peaks were observed in LSV curves, indicating that the self-reduction of $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$ occurred during the LSV test. Both $SnS_{0.09}/CC$ and ${\sf SnS}_{0.55}/{\sf CC}$ exhibited higher current densities in ${\sf CO}_{2}$ - than in N_2 -saturated electrolyte, implying their promising applications in $CO₂$ electroreduction. Furthermore, SnS_{0.55}/CC exhibited a larger current density than that of SnS $_{0.09}$ /CC in CO $_2$ -saturated electrolyte, suggesting $SnS_{0.55}/CC$ was more active for electrocatalytic CO₂RR. At potentials of −1.0 V and −1.1 V vs. RHE, $\text{SnS}_{0.55}/\text{CC}$ exhibited considerable current densities of 21.8 mA/cm² and 30.5 mA/cm², respectively. To further evaluate $CO₂RR$ activity and selectivity, chronoamperometry tests were conducted at different potentials. Quantified by gas chromatography and ion chromatography analysis, the catalytic products in $CO₂$ electroreduction included formate (HCOO⁻), CO, and H₂ over the SnS_{0.09}/CC and SnS_{0.55}/CC. As shown in [Fig. 3b](#page-4-1) and S11, formate was the major product over the whole potential range, while CO and H_2 were the minor products for the $\text{SnS}_{0.09}/\text{CC}$ and $\text{SnS}_{0.55}/\text{CC}$. $\text{SnS}_{0.55}/\text{CC}$ exhibited higher selectivity to formate, achieving a formate FE of 93.1% at −1.1 V vs. RHE, whereas the maximum formate FE of $SnS_{0.09}/CC$ was 86.8% [\(Fig. 3b](#page-4-1), S11 and 12c). Compared with $\text{SnS}_{0.55}/\text{CC}$, the lower selectivity for formate of $\text{SnS}_{0.09}/\text{CC}$ was attributed to its preference for hydrogen evolution reactions [\(Fig. 3b](#page-4-1), S11 and 12a). There was no significant difference in FE for CO production between the $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$ [\(Fig. 3b](#page-4-1), S11, 12a and S12b). We further calculated the partial current density (J) of the $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$ at all of applied potentials (Fig. S13). At −1.1 V vs. RHE, $SnS_{0.55}/CC$ exhibited a J_{HCOO}- of 28.4 mA/cm², [which](#page-4-1) was lar-ger than that of SnS_{0.09}/CC (22.1 mA/cm², [Fig. 3c](#page-4-1)). J_{H2} of $SnS_{0.55}/CC$ was slightly less than that of $SnS_{0.09}/CC$, and J_{CO} of $SnS_{0.55}/CC$ was slightly higher than that of $SnS_{0.09}/CC$ (Fig. S14). The stability of the $\text{SnS}_{0.09}/\text{CC}$ and $\text{SnS}_{0.55}/\text{CC}$ in CO_2RR were examined at a constant applied potential of -1.1 V vs. RHE for 12 h (Fig. S15). Both $\text{SnS}_{0.09}/\text{CC}$ and $\text{SnS}_{0.55}/\text{CC}$ exhibited steady FEs for formate of about 83.3% and 90.4%, respectively. The current densities of both electrodes showed a trend of increasing with time, in which the current densities increasing more rapidly in the first 2 hours. After 12 hours of stability testing, both electrodes became whiter in color and the particle sizes became larger (Fig. S16 and S17). The EDS and the XPS quantitative analysis results reflected that after 12 hours of reaction, S content among the two electrodes got decreased (Fig. S16g and S18-S20), and similar Sn/S atom ratio was found for $SnS_{0.09}/CC-12$ h and $SnS_{0.55}/CC-12$ h (Table S3 and S4). XRD patterns and the Sn 3*d* XPS spectra of

Fig. 3 a, Geometrical current densities over the SnS_{0.09}/CC and SnS_{0.55}/CC in CO₂- and N₂-saturated 0.5 M KHCO₃. **b**, Potential-dependent FEs of CO, H₂ and formate and **c**, partial current densities of formate (J_{HCOO}−) over the SnS_{0.09}/CC and SnS_{0.55}/CC. **d**, Tafel plots of the SnS_{0.09}/CC and SnS_{0.55}/CC. **e**, Charging current density differences plotted against scan rates for the SnS_{0.09}/CC and SnS_{0.55}/CC. The fitting slopes are twice that of the Cdl values. **f**, Nyquist plots of the $S_{0.09}/CC$ and $S_{0.55}/CC$ at -0.8 V vs. RHE.

 $SnS_{0.09}/CC-12$ h demonstrated that $SnS_{0.09}/CC$ was reduced to Sn (Fig. S16e and S16f). Compared with $\text{SnS}_{0.55}/\text{CC}}$, SnS $_{0.55}/\text{CC}$ -12 h after 12-h stability test showed clear XRD diffraction peaks, but no obvious characteristic peak for metallic Sn was observed in the XPS spectra. The above characterization results indicated that the electrocatalysts continued to undergo in-situ transformation under the action of a long-time CO₂ reduction potential. Compared with other tin sulfide materials, $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$ had comparable electrocatalytic properties and milder synthesis conditions (Table S5).

To further understand the different activities of $SnS_{0.09}/CC$ and $\text{SnS}_{0.55}/\text{CC}$ in CO_2 electroreduction, we investigated the relevant reaction kinetics. [Fig. 3d](#page-4-1) showed the Tafel plots (overpotentials for formate production vs. log J_{HCOO}-) of the two electrodes. The Tafel slopes of $\text{SnS}_{0.09}/\text{CC}$ and $\text{SnS}_{0.55}/\text{CC}$ were calculated to be 141.3 and 120.9 mV/dec, respectively. Theoretically, the Tafel slope of activation process is 118 mV/dec. The Tafel slopes of the two electrodes were both close to 118 mV/dec, proving that the activation of CO $_2$ serves as the rate-limiting step in the system of SnS_x.^{[[30](#page-6-10)[,37,](#page-6-13)[43\]](#page-6-17)} The Cdl values of $SnS_{0.55}/CC$ (12.3 mF/cm²) was slightly larger than that of SnS $_{0.09}$ /CC (10.3 mF/cm²) (Fig. S21). Given that the Cdl value is positively correlated with the electrochemical active surface area (ECSA), the geometrical current density was normalized by Cdl values to evaluate the intrinsic activity of catalysts. The Cdl-normalized current density demonstrated that the different ECSA of the two catalysts brought by different potential regions of CV activation was not the main reason for the enhanced current density (Fig. S22). Further, [Fig. 3f](#page-4-1) showed the Nyquist plots of the $SnS_{0.09}/CC$ and $SnS_{0.55}/CC$. The charge transfer resistance of $S_{0.55}/CC$ was smaller than that of $SnS_{0.09}/CC$, indicating a faster reaction rate.^{[[30](#page-6-10),[37](#page-6-13)[,43\]](#page-6-17)} Based on the above analysis, $SnS_{0.55}/CC$ promoted the kinetics of $CO₂$ activation process in $CO₂$ electroreduction.

Discussion

It's found that the composition and morphology of SnS_x chalcogel were affected by the regulation of CV activation, thus resulting in different performances for electrocatalytic $CO₂RR$. Using SnS_x as a pre-catalyst, SnS_{0.09}/CC and SnS_{0.55}/CC were obtained through CV activation in different potential ranges. According to the characterizations, metal Sn was generated on the $SnS_{0.09}/CC$ electrodes after CV activation. $SnS_{0.09}/CC$ had less S contents than $SnS_{0.55}/CC$. Both electrodes showed high selectivity to formate in the electrocatalytic CO₂RR. SnS_{0.55}/CC had a higher current density and FE_{H-} coo⁻ than SnS_{0.09}/CC. This phenomenon could be attributed to the different S content in $\text{SnS}_{0.09}/\text{CC}$ and $\text{SnS}_{0.55}/\text{CC}$ obtained by CV activation at different potential intervals. Being different from $SnS_{0.09}/CC$, the formation for $SnS_{0.55}/CC$ underwent oxidation reactions with S ions, resulting in higher S content in $\text{SnS}_{0.55}/\text{CC}$. Different degrees of the peaks shift of Sn and S in XPS spectra and the reaction kinetics analysis indicated that the presence of more S affects the electronic structure of the catalysts and promotes the kinetics of $CO₂$ activation. There were also works that demonstrated that the presence of S enhanced the formation of formate.^{[[46](#page-6-20),[49](#page-6-23)[,50\]](#page-6-24)} The presence of S could weaken the HCOO* adsorption energies, thus fa-voring the formation of HCOO* toward formate.^{[[47](#page-6-21)[,49\]](#page-6-23)}, In our work, although both electrodes still underwent transformation after 12-h stability test, FE_{HCOO}- could be maintained and $SnS_{0.55}/CC$ outperformed $SnS_{0.09}/CC$. Our work sheds light on the effect of the CV activation process for the construction of "real" electrocatalysts for $CO₂RR$.

Conclusions

In conclusion, tin sulfide chalcogel was prepared by a simple and mild metathesis reaction and applied as the precatalyst for electrochemical $CO₂RR$. Tin sulfide chalcogel derived catalysts with different S contents were obtained through CV activation at different potential intervals, as the electrocatalysts underwent different redox processes. $SnS_{0.55}/CC$ with higher S contents exhibited higher current density and a higher Faraday efficiency for formate than $SnS_{0.09}/CC.$ The experimental results suggested that higher S content affected the interaction between the catalysts and $CO₂$, promoted $CO₂$ activation and favored the formation of formate. Our work revealed that different S contents of catalysts affected the performance of electrocatalytic $CO₂RR$, while the influence of the choice of potential intervals in CV activation on the electrocatalysts conversion also brought valuable insights for other electrocatalytic applications.

■ ACKNOWLEDGMENTS

This research was financially supported by the Natural Science Foundation of Shandong Province (ZR2019JQ05, ZR2019ZD47), the Education Department of Shandong Province (2019KJC006) and the National Natural Science Foundation of China (22175104).

■ CONFLICT OF INTEREST

The authors declare no conflict of interest.

■ AUTHOR CONTRIBUTIONS

J. Liu conceived the idea and designed the experiments. J. Bai, L. Yang and J. Liu cowrote the paper. J. Bai synthesized and characterized the samples and further analyzed the results. J. Bai performed the catalytic tests. All authors discussed and commented on the manuscript.

■ REFERENCES

- S. Bag, P. N. Trikalitis, P. J. Chupas, G. S. Armatas and M. G. Kanatzidis, *[Science](https://doi.org/10.1126/science.1142535)*, 2007, 317, 490 1.
- S. Bag, I. U. Arachchige and M. G. Kanatzidis, *[J. Mater. Chem.](https://doi.org/10.1039/b804011g)*, 2008, 18, 3628 2.
- S. Bag, A. F. Gaudette, M. E. Bussell and M. G. Kanatzidis, *[Nat.](https://doi.org/10.1038/nchem.208) [Chem.](https://doi.org/10.1038/nchem.208)*, 2009, 1, 217 3.
- 4. Y. Oh, S. Bag, C. D. Malliakas and M. G. Kanatzidis, *[Chem. Mater.](https://doi.org/10.1021/cm2003462)*, 2011, 23, 2447
- K. Polychronopoulou, C. D. Malliakas, J. He and M. G. Kanatzidis, *[Chem. Mater.](https://doi.org/10.1021/cm301444p)*, 2012, 24, 3380 5.
- S. Chandrasekaran, L. Yao, L. Deng, C. Bowen, Y. Zhang, S. Chen, Z. Lin, F. Peng and P. Zhang, *[Chem. Soc. Rev.](https://doi.org/10.1039/C8CS00664D)*, 2019, 48, 4178 6.
- Y. Zhang, Y. Zhang, H. Zhang, L. Bai, L. Hao, T. Ma and H. Huang, *[Coord. Chem. Rev.](https://doi.org/10.1016/j.ccr.2021.214147)*, 2021, 448, 214147 7.
- K. S. Subrahmanyam, D. Sarma, C. D. Malliakas, K. Polychronopoulou, B. J. Riley, D. A. Pierce, J. Chun and M. G. Kanatzidis, 8.

[Chem. Mater.](https://doi.org/10.1021/acs.chemmater.5b00413), 2015, 27, 2619

- Y. Shim, R. M. Young, A. P. Douvalis, S. M. Dyar, B. D. Yuhas, T. Bakas, M. R. Wasielewski and M. G. Kanatzidis, *[J. Am. Chem. Soc.](https://doi.org/10.1021/ja507297p)*, 2014, 136, 13371 9.
- 10. K. S. Subrahmanyam, C. D. Malliakas, S. M. Islam, D. Sarma, J. Wu and M. G. Kanatzidis, *[Chem. Mater.](https://doi.org/10.1021/acs.chemmater.6b02913)*, 2016, 28, 7744
- 11. B. D. Yuhas, C. Prasittichai, J. T. Hupp and M. G. Kanatzidis, [J. Am.](https://doi.org/10.1021/ja205981v) *[Chem. Soc.](https://doi.org/10.1021/ja205981v)*, 2011, 133, 15854
- 12. A. Banerjee, B. D. Yuhas, E. A. Margulies, Y. Zhang, Y. Shim, M. R. Wasielewski and M. G. Kanatzidis, *[J. Am. Chem. Soc.](https://doi.org/10.1021/ja512491v)*, 2015, 137, 2030
- 13. M. Shafaei-Fallah, A. Rothenberger, A. P. Katsoulidis, J. He, C. D. Malliakas and M. G. Kanatzidis, *[Adv. Mater.](https://doi.org/10.1002/adma.201102006)*, 2011, 23, 4857
- 14. K. S. Subrahmanyam, I. Spanopoulos, J. H. Chun, B. J. Riley, P. K. Thallapally, P. N. Trikalitis and M. G. Kanatzidis, *[ACS Appl. Mater.](https://doi.org/10.1021/acsami.6b15896) [Interfaces](https://doi.org/10.1021/acsami.6b15896)*, 2017, 9, 33389
- 15. Z. Hassanzadeh Fard, S. M. Islam and M. G. Kanatzidis, *[Chem. Ma](https://doi.org/10.1021/acs.chemmater.5b02805)[ter.](https://doi.org/10.1021/acs.chemmater.5b02805)*, 2015, 27, 6189
- 16. B. J. Riley, J. Chun, W. Um, W. C. Lepry, J. Matyas, M. J. Olszta, X. Li, K. Polychronopoulou and M. G. Kanatzidis, *[Environ. Sci. Technol.](https://doi.org/10.1021/es400595z)*, 2013, 47, 7540
- 17. J. Liu, K. He, W. Wu, T. B. Song and M. G. Kanatzidis, [J. Am. Chem.](https://doi.org/10.1021/jacs.6b13279) *[Soc.](https://doi.org/10.1021/jacs.6b13279)*, 2017, 139, 2900
- 18. B. D. Yuhas, A. L. Smeigh, A. P. Samuel, Y. Shim, S. Bag, A. P. Douvalis, M. R. Wasielewski and M. G. Kanatzidis, *[J. Am. Chem.](https://doi.org/10.1021/ja111275t) [Soc.](https://doi.org/10.1021/ja111275t)*, 2011, 133, 7252
- 19. J. Staszak-Jirkovsky, C. D. Malliakas, P. P. Lopes, N. Danilovic, S. S. Kota, K. C. Chang, B. Genorio, D. Strmcnik, V. R. Stamenkovic, M. G. Kanatzidis and N. M. Markovic, *[Nat. Mater.](https://doi.org/10.1038/nmat4481)*, 2016, 15, 197
- 20. X. Shan, J. Liu, H. Mu, Y. Xiao, B. Mei, W. Liu, G. Lin, Z. Jiang, L. Wen and L. Jiang, *[Angew Chem. Int. Ed.](https://doi.org/10.1002/anie.201911617)*, 2020, 59, 1659
- H. Mu, G. Lin, Y. Zhang, Y. Xiao and J. Liu, *[Colloids Surf. A Physico-](https://doi.org/10.1016/j.colsurfa.2021.126734)*21. *[chem. Eng. Asp.](https://doi.org/10.1016/j.colsurfa.2021.126734)*, 2021, 623, 126734
- 22. G. Wang, J. Chen, Y. Ding, P. Cai, L. Yi, Y. Li, C. Tu, Y. Hou, Z. Wen and L. Dai, *[Chem. Soc. Rev.](https://doi.org/10.1039/D0CS00071J)*, 2021, 50, 4993
- 23. Y. Y. Birdja, E. Pérez-Gallent, M. C. Figueiredo, A. J. Göttle, F. Calle-Vallejo and M. T. M. Koper, *[Nat. Energy](https://doi.org/10.1038/s41560-019-0450-y)*, 2019, 4, 732
- M. G. Kibria, J. P. Edwards, C. M. Gabardo, C. T. Dinh, A. Seifi-24. tokaldani, D. Sinton and E. H. Sargent, *[Adv. Mater.](https://doi.org/10.1002/adma.201807166)*, 2019, 31, 1807166
- 25. S. A. Patil, H. T. Bui, S. Hussain, I. Rabani, Y. Seo, J. Jung, N. K. Shrestha, H. Kim and H. Im, *[Dalton Trans.](https://doi.org/10.1039/D1DT01855H)*, 2021, 50, 12723
- 26. S. A. Patil, N. K. Shrestha, A. I. Inamdar, C. Bathula, J. Jung, H. Im and H. Kim, *[Int. J. Energy Res.](http://dx.doi.org/10.1002/er.8212),* 2022
- 27. S. A. Patil, N. K. Shrestha, S. Hussain, J. Jung, S. W. Lee, C. Bathula, A. N. Kadam, H. Im and H. Kim, *[J. Hazard. Mater.](https://doi.org/10.1016/j.jhazmat.2021.126105)*, 2021, 417, 126105
- W. Yu, F. Shu, Y. Huang, F. Yang, Q. Meng, Z. Zou, J. Wang, Z. 28. Zeng, G. Zou and S. Deng, *[J. Mater. Chem. A](https://doi.org/10.1039/D0TA06576E)*, 2020, 8, 20677
- 29. M. Chen, S. Wan, L. Zhong, D. Liu, H. Yang, C. Li, Z. Huang, C. Liu, J. Chen, H. Pan, D. S. Li, S. Li, Q. Yan and B. Liu, *[Angew Chem. Int.](https://doi.org/10.1002/anie.202111905) [Ed.](https://doi.org/10.1002/anie.202111905)*, 2021, 60, 26233
- 30. A. Zhang, Y. Liang, H. Li, S. Wang, Q. Chang, K. Peng, Z. Geng and J. Zeng, *[Nano Lett.](https://doi.org/10.1021/acs.nanolett.1c02757)*, 2021, 21, 7789
- S. Zhao, S. Li, T. Guo, S. Zhang, J. Wang, Y. Wu and Y. Chen, *[Nano-](https://doi.org/10.1007/s40820-019-0293-x)*31. *[Micro Lett.](https://doi.org/10.1007/s40820-019-0293-x)*, 2019, 11, 62
- F. Wei, T. Wang, X. Jiang, Y. Ai, A. Cui, J. Cui, J. Fu, J. Cheng, L. Lei, 32. Y. Hou and S. Liu, *[Adv. Funct. Mater.](https://doi.org/10.1002/adfm.202002092)*, 2020, 30, 2002092
- W. Ni, Y. Gao, Y. Lin, C. Ma, X. Guo, S. Wang and S. Zhang, *[ACS](https://doi.org/10.1021/acscatal.0c05514)* 33. *[Catal.](https://doi.org/10.1021/acscatal.0c05514)*, 2021, 11, 5212
- 34. Z. Wu, H. Wu, W. Cai, Z. Wen, B. Jia, L. Wang, W. Jin and T. Ma, [An](https://doi.org/10.1002/anie.202102832)*[gew Chem. Int. Ed.](https://doi.org/10.1002/anie.202102832)*, 2021, 60, 12554
- 35. Y. J. Ko, J. Y. Kim, W. H. Lee, M. G. Kim, T. Y. Seong, J. Park, Y. Jeong, B. K. Min, W. S. Lee, D. K. Lee and H. S. Oh, *[Nat. Commun.](https://doi.org/10.1038/s41467-022-29783-7)*, 2022, 13, 2205
- 36. X. Zhang, M. Jiao, Z. Chen, X. Ma, Z. Wang, N. Wang, X. Zhang and L. Liu, *[Chem. Eng. J.](https://doi.org/10.1016/j.cej.2021.132145)*, 2022, 429, 132145
- 37. Z. Chen, X. Zhang, M. Jiao, K. Mou, X. Zhang and L. Liu, [Adv. En](https://doi.org/10.1002/aenm.201903664)*[ergy Mater.](https://doi.org/10.1002/aenm.201903664)*, 2020, 10, 1903664
- G. A. El-Nagar, A. M. Mohammad, M. S. El-Deab and B. E. El-38. Anadouli, *[ACS Appl. Mater. Interfaces](https://doi.org/10.1021/acsami.7b01565)*, 2017, 9, 19766
- 39. F. Joo, *[ChemSusChem](https://doi.org/10.1002/cssc.200800133)*, 2008, 1, 805
- 40. Q.-L. Zhu and Q. Xu, *[Energy Environ. Sci.](https://doi.org/10.1039/C4EE03690E)*, 2015, 8, 478
- 41. Q. Gong, P. Ding, M. Xu, X. Zhu, M. Wang, J. Deng, Q. Ma, N. Han, Y. Zhu, J. Lu, Z. Feng, Y. Li, W. Zhou and Y. Li, *[Nat. Commun.](https://doi.org/10.1038/s41467-019-10819-4)*, 2019, 10, 2807
- 42. S. Bag and M. G. Kanatzidis, *[J. Am. Chem. Soc.](https://doi.org/10.1021/ja1059284)*, 2010, 132, 14951
- 43. A. Zhang, R. He, H. Li, Y. Chen, T. Kong, K. Li, H. Ju, J. Zhu, W. Zhu and J. Zeng, *[Angew Chem. Int. Ed.](https://doi.org/10.1002/anie.201806043)*, 2018, 57, 10954
- F. Li, L. Chen, M. Xue, T. Williams, Y. Zhang, D. R. MacFarlane and 44. J. Zhang, *[Nano Energy](https://doi.org/10.1016/j.nanoen.2016.11.004)*, 2017, 31, 270
- 45. B. Ni, T. He, J. O. Wang, S. Zhang, C. Ouyang, Y. Long, J. Zhuang and X. Wang, *[Chem. Sci.](https://doi.org/10.1039/C7SC05452A)*, 2018, 9, 2762
- 46. X. Zheng, P. De Luna, F. P. García de Arquer, B. Zhang, N. Becknell, M. B. Ross, Y. Li, M. N. Banis, Y. Li, M. Liu, O. Voznyy, C. T. Dinh, T. Zhuang, P. Stadler, Y. Cui, X. Du, P. Yang and E. H. Sargent, *[Joule](https://doi.org/10.1016/j.joule.2017.09.014)*, 2017, 1, 794
- 47. T. Shinagawa, G. O. Larrazábal, A. J. Martín, F. Krumeich and J. Pérez-Ramírez, *[ACS Catal.](https://doi.org/10.1021/acscatal.7b03161)*, 2018, 8, 837
- 48. Y. Zhang, L. Hu and W. Han, *[J. Mater. Chem. A](https://doi.org/10.1039/C8TA08698B)*, 2018, 6, 23610
- Y. Deng, Y. Huang, D. Ren, A. D. Handoko, Z. W. Seh, P. Hirunsit 49. and B. S. Yeo, *[ACS Appl. Mater. Interfaces](https://doi.org/10.1021/acsami.8b08428)*, 2018, 10, 28572
- W. Ma, S. Xie, X. G. Zhang, F. Sun, J. Kang, Z. Jiang, Q. Zhang, D. Y. 50. Wu and Y. Wang, *[Nat. Commun.](https://doi.org/10.1038/s41467-019-08805-x)*, 2019, 10, 892

\odot $|$ (cc)

©2023 The Authors. *Materials Lab* is published by Lab Academic Press. This is an open access article under the terms of the Creative Commons Attribution License, which permits use, distribution and reproduction in any medium, provided the original work is properly cited.